

Challenges and Prospects of Lithium—Sulfur Batteries

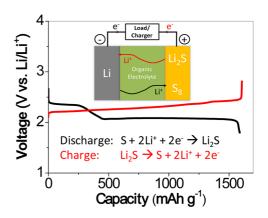
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CONSPECTUS

E lectrical energy storage is one of the most critical needs of 21st century society. Applications that depend on electrical energy storage include portable electronics, electric vehicles, and devices for renewable energy storage from solar and wind. Lithium-ion (Li-ion) batteries have the highest energy density among the rechargeable battery chemistries. As a result, Li-ion batteries have proven successful in the portable electronics market and will play a significant role in large-scale energy storage. Over the past two decades, Li-ion batteries based on insertion cathodes have reached a cathode capacity of \sim 250 mA h g $^{-1}$ and an energy density of \sim 800 W h kg $^{-1}$, which do not meet the requirement of \sim 500 km between charges for all-electric vehicles. With a goal of increasing energy density, researchers are pursuing alternative cathode materials such as sulfur and O_2 that can offer capacities that exceed those of conventional insertion



cathodes, such as $LiCoO_2$ and $LiMn_2O_4$, by an order of magnitude (>1500 mA h g⁻¹). Sulfur, one of the most abundant elements on earth, is an electrochemically active material that can accept up to two electrons per atom at \sim 2.1 V vs Li/Li^+ . As a result, sulfur cathode materials have a high theoretical capacity of 1675 mA h g⁻¹, and lithium—sulfur (Li-S) batteries have a theoretical energy density of \sim 2600 W h kg⁻¹. Unlike conventional insertion cathode materials, sulfur undergoes a series of compositional and structural changes during cycling, which involve soluble polysulfides and insoluble sulfides. As a result, researchers have struggled with the maintenance of a stable electrode structure, full utilization of the active material, and sufficient cycle life with good system efficiency. Although researchers have made significant progress on rechargeable Li-S batteries in the last decade, these cycle life and efficiency problems prevent their use in commercial cells.

To overcome these persistent problems, researchers will need new sulfur composite cathodes with favorable properties and performance and new Li—S cell configurations. In this Account, we first focus on the development of novel composite cathode materials including sulfur—carbon and sulfur—polymer composites, describing the design principles, structure and properties, and electrochemical performances of these new materials. We then cover new cell configurations with carbon interlayers and Li/dissolved polysulfide cells, emphasizing the potential of these approaches to advance capacity retention and system efficiency. Finally, we provide a brief survey of efficient electrolytes. The Account summarizes improvements that could bring Li—S technology closer to mass commercialization.

1. Introduction

Lithium-ion (Li-ion) batteries have dominated the battery market for portable electronics since their launch in the early 1990s by Sony and are believed to be promising candidates for electric vehicles and grid energy storage due to their higher energy density compared with other battery chemistries.^{1–4} Insertion cathode materials, such as LiCoO₂, LiMn₂O₄, and LiFePO₄, have been well developed in the last 2

decades and are being widely used in current Li-ion batteries. 2,5 To improve the energy density of Li-ion batteries further to meet the requirement of ~ 500 km range between charges for all-electric vehicles and grid energy storage, alternative cathodes with a significant reduction in cost and increase in capacity and energy density are needed. 6

Sulfur, one of the most abundant elements on earth, is a promising cathode with the highest theoretical capacity of

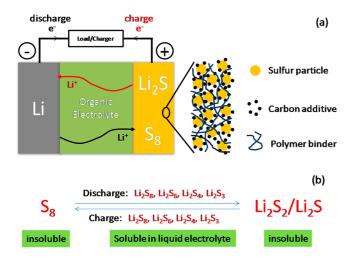


FIGURE 1. (a) Illustration of the charge (red)/discharge (black) process involved in a rechargeable Li-S cell consisting of lithium metal anode, organic electrolyte, and sulfur composite cathode and (b) charge/discharge process involving the formation of soluble lithium polysulfides (Li $_2$ S $_8$, Li $_2$ S $_6$, Li $_2$ S $_4$, and Li $_2$ S $_3$) and insoluble Li $_2$ S $_2$ /Li $_2$ S.

1675 mA h g⁻¹ among the solid elements. Moving from the traditional insertion cathodes to sulfur has many benefits besides the high capacity, such as the low operating voltage (2.15 V vs Li/Li⁺) improving the safety, the low-cost sulfur replacing expensive transition metals like cobalt, and environmentally friendly sulfur compared with certain toxic transition-metal compounds. An order of magnitude higher capacity than that of the conventional insertion compound cathodes can enable packaged lithium-sulfur (Li-S) cells with an energy density of 400–600 W h kg⁻¹, which is two or three times higher than that of current Li-ion batteries, and it could meet the target for the 500 km range.⁶ The low-cost and high energy density are also appealing for grid energy storage for renewable energies such as solar and wind if long cycle life and high system efficiency can be achieved.

Conventional Li—S cells consist of a lithium metal anode, an organic liquid electrolyte, and a sulfur composite cathode (Figure 1a). There are several major issues facing rechargeable Li—S batteries that impede their practical applications. Sulfur is in the form of polyatomic molecules with different structures, and octasulfur (cyclo-S₈) is the most stable allotrope at room temperature. Cyclo-S₈ undergoes a series of structural and morphological changes during the charge—discharge process involving the formation of soluble lithium polysulfides Li_2S_x ($8 \le x \le 3$) and insoluble sulfides $\text{Li}_2\text{S}_2/\text{Li}_2\text{S}_2$ in liquid electrolyte (Figure 1b). The high resistances of sulfur and the intermediates formed along with their structural changes result in unstable electrochemical contact within sulfur electrodes. In addition, the dissolved polysulfides

shuttle between the anode and cathode during the charge process involving side reduction reactions with lithium anode and reoxidation reactions at the cathode. These issues result in a low utilization of active material, poor cycle life, and low system efficiency. Apparently, it is impossible to meet all the requirements with the conventional Li—S battery electrode formula consisting of micrometer-sized active material, carbon, and polymer binder (Figure 1a). A variety of strategies have been developed to form sulfur composites with favorable structures and properties to improve the discharge capacity, cyclability, and Coulombic efficiency. Other approaches being pursued include novel cell configurations with trapping interlayers, Li/dissolved polysulfide cells, and efficient electrolytes. 9–12

2. Sulfur—Carbon Composite Cathodes

2.1. Porous Carbon-Based Composites. The main efforts to improve the electrochemical performances of Li—S batteries have been on the development of carbon-based sulfur composites. Following the report of high-capacity Li—S cells in 2009 with highly ordered mesoporous carbon and sulfur, studies on sulfur—carbon composite cathodes have grown rapidly. An ideal carbon matrix for sulfur—carbon composites needs to have (i) high electrical conductivity, (ii) electrochemical affinity for sulfur, (iii) small cells (pores) without large outlets to accommodate polysulfides, (iv) accessibility of liquid electrolyte to active material, and (v) stable framework to sustain the strain generated by the volume changes of the active material during cycling.

Recent progress on the sulfur-carbon composites benefits from their hierarchical design (Figure 2). Carbon spheres with micropores can encapsulate sulfur (Figure 2a), but the total pore volume needs to be increased to enhance the sulfur loading.¹⁴ Spherical ordered mesoporous carbon with a large number of inner pores is a superior option to serve as the matrix to accommodate more sulfur without compromising battery performance (Figure 2b).¹⁵ Porous hollow carbon-sulfur composites exhibit impressive cyclability over 100 cycles (Figure 2c). 16 Graphene oxide has also been used (Figure 2d), but the graphene-sulfur composite may need further treatment for immobilizing the intermediate polysulfides. 17,18 Porous carbon nanofibers and hollow carbon nanofibers or nanotubes are also good substrates to be impregnated with sulfur (Figure 2e,f). 19-21 Although these carbon-based sulfur composites have shown outstanding progress in Li-S battery performance, high active material loading (>70 wt %), high specific capacity (>1200 mA h g^{-1}), and excellent cycle stability and life

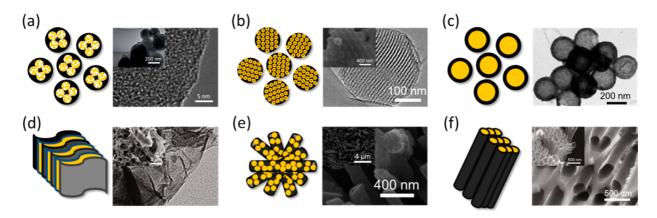


FIGURE 2. Hierarchical designs of carbon-based sulfur composites: (a) microporous carbon spheres, (b) spherical ordered mesoporous carbon nanoparticles, (c) porous hollow carbon, (d) graphene oxide sheets, (e) porous carbon nanofibers, and (f) hollow carbon nanofibers to encapsulate sulfur. Panels a and e were reproduced with permission from refs 14 and 19. Copyright 2010 and 2011 Royal Society of Chemistry. Panels b and c were reproduced with permission from refs 15 and 16. Copyright 2012 and 2011 Wiley. Panels d and f were reprinted with permission from refs 17 and 20. Copyright 2011 American Chemical Society.

TABLE 1. Summary of Representative Synthesis Routes To Make Sulfur—Carbon Composite Cathode Materials

method	description	binding between S and C	feature	ref
mixing ball-milling thermal treatment	mixing materials by magnetic stirring ball-milling materials with binders one step heating: 155 °C two step heating: 150–200 °C; 300–350 °C	weak medium strong strong	high-energy ball-milling good for mesoporous carbons superficial sulfur removed by the	24 22 13, 27 18, 25, 26
heterogeneous nucleation	sulfur vaporizing $S_x^{2-} + 2H^+ \rightarrow (x-1)S_1^{\downarrow} + H_2S$ $SO_2 + 2S_2^{2-} + 4H^+ \rightarrow 3S_1^{\downarrow} + 2H_2O$ $S_2O_3^{2-} + 2H^+ \rightarrow S_1^{\downarrow} + H_2SO_3$	very strong strong strong strong	second step molecular infiltration producing hazardous H ₂ S gas involving the fewest chemicals	16 19, 28 29 30

(<10% capacity loss over 100 cycles) could not be obtained simultaneously with any single composite material.

2.2. Composites Synthesized by Various Methods. In conventional sulfur electrodes, ball-milling or mixing sulfur powder and carbon materials have been used.²²⁻²⁴ The poor contact between sulfur and carbon results in low initial discharge capacities (e.g., 300-500 mA h g⁻¹), so improved processes to prepare sulfur-carbon composites have been developed (Table 1). For example, a two-step sulfur melting route was first reported by Wang et al., 25 involving a thermal treatment at 200 °C to melt sulfur that can diffuse into the pores of active carbon followed by a second firing at 300 °C to vaporize sulfur covering the outer surface of active carbon. A high first discharge capacity of 800 mA h g⁻¹ was achieved with this composite. This heating process has been further used with slightly modified temperature and dwell time and with different carbon materials to obtain high performance sulfur-carbon composites. 18,26 One-step sulfur melting route was later employed especially with mesoporous carbons, which could effectively absorb fused sulfur into the interconnected pore structure during synthesis without further thermal treatment. 13,27 Jayaprakash et al. 16

utilized a sulfur vaporizing route to infuse gaseous sulfur into porous hollow carbon and obtained outstanding cyclability as sulfur vapor led to molecular contacts with the carbon framework.

Heterogeneous sulfur nucleation on carbon matrices in an aqueous solution has become a favorable process not only because of the low-cost solution-based reactions but also due to the elimination of toxic solvents. Sulfur nucleates on dispersed carbon substrates after dropping acid into the polysulfide solution prepared by the dissolution of sulfur in a sodium sulfide (Na₂S) solution. 19,28 Another method involving Na₂SO₃, Na₂S, and acid has led to a core—shell carbon/ sulfur material.²⁹ An in situ sulfur deposition route has also been developed to synthesize carbon-wrapped sulfur composites.³⁰ High-purity sulfur heterogeneously nucleates on highly dispersed carbon nanoparticles in an aqueous solution as the thiosulfate ions are reduced or sulfide ions are oxidized to sulfur. These synthesis methods involve a simple water-based reaction to bind sulfur and carbon strongly and are applicable to any carbon substrate.

2.3. Binder-Free Composite Electrodes. Owing to establishing green processes, avoiding toxic chemicals in the

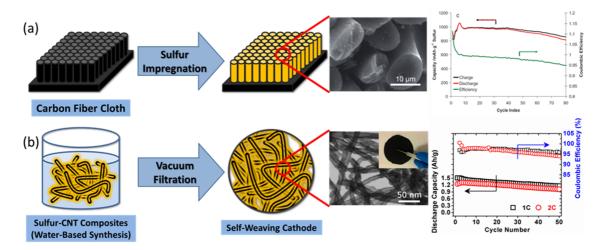


FIGURE 3. Comparison of binder-free cathodes: (a) Carbon substrate-based composite cathode Reproduced with permission from ref 32. Copyright 2011 Wiley. (b) Self-weaving sulfur—CNT composite cathode.

manufacturing of battery electrodes is appealing. *N*-Methyl-2-pyrrolidone (NMP) is a solvent generally used in fabricating Li-ion battery electrodes to disperse the polymer binder and active materials. However, NMP is harmful to human health and the environment.³¹ One possible approach is making binder-free electrodes, which can eliminate the use of NMP. By removing the binder, the active material loading and electrode conductivity can also be increased.

Two methods for preparing binder-free sulfur composite cathodes have been developed (Figure 3). Aurbach et al.³² have fabricated a sulfur-impregnated activated carbon fiber cloth as a binder-free cathode (Figure 3a). They employed a commercial carbon cloth as the substrate, and sulfur was injected into the pores of carbon fibers by cofiring sulfur and the carbon cloth. Vertically aligned carbon nanotubes grown on a metal foil can be used as a binder-free cathode matrix as well, which is then infiltrated with a sulfur solution and melt.³³ Both these binder-free features of the cathodes were inherited from the formed robust substrates. Another approach was depositing sulfur onto multiwalled carbon nanotubes (MWCNT) via an in situ sulfur deposition route, and the sulfur-carbon nanotube composite forms a free-standing paper after vacuum filtration without binders (Figure 3b). Curly carbon nanotubes were specifically used to have a selfintertwining behavior.34,35 This binder-free sulfur-carbon nanotube composite shows excellent high-rate cycle performance, retaining >1000 mA h g⁻¹ for 50 cycles even at 2 C rate (Figure 3b).³⁶ The high capacity in the composite at high rates is attributed to the superior electrical properties of MWCNT, resulting in low resistance in the cathode. The tortuous interspaces located in the network of MWCNT offer a stable accommodation of sulfur for rapid redox reactions

TABLE 2. Summary of the Voltage Window and Electrode Performance of Possible Insertion Hosts for Hybrid Sulfur Composite Cathodes

electrode material	reduced product	voltage (V)	capacity (mA h g^{-1})	ref
S ₈	Li ₂ S	3.0-1.5	>1000	
TiS ₂	Li _x TiS ₂	3.0 - 1.5	240	39
VO ₂ (B)	$Li_{1-x}VO_2$	3.5 - 2.0	200	41
Li ₄ Ti ₅ O ₁₂	Li ₇ Ti ₅ O ₁₂	2.0 - 1.2	150	42
TiO ₂ (B)	$Li_{1-x}TiO_2$	2.5 - 1.0	200	45
$Li_2C_6O_6$	$Li_4C_6O_6$	2.5 - 1.5	200	46

and prevent sulfur/Li₂S from forming extensive agglomerates during cycling, leading to good cyclability. In addition, the composite cathode is also current collector-free, further reducing the net weight of cathodes.

2.4. Hybrid Sulfur Composite Cathodes. Although sulfur-carbon composites have shown good cyclability, about 30-50% of the electrode weight belongs to inactive carbon and binder. To improve the energy density, the active material loading in the cathode must be increased. A prospective strategy is replacing a portion of carbon and binder by a lithium insertion oxide or sulfide. These materials need to work in the voltage range similar to that of sulfur (3-1.5 V)to contribute capacities. Some potential candidates are summarized in Table 2. TiS₂ operates in exactly the same voltage window as sulfur. $^{37-39}$ Nanosized VO_2-B with a sloping discharge profile can be partially lithiated by controlling the terminal operating potentials. 40,41 Nanostructured spinel $\text{Li}_4\text{Ti}_5\text{O}_{12}^{42,43}$ and $\text{TiO}_2\text{--}\text{B}^{44,45}$ are other possibilities. Furthermore, there is growing interest in organic redox hosts such as Li₄C₆O₆, 46 which is another potential candidate. These lithium insertion hosts could be used as a matrix with reduced amount of carbon to generate sulfur by the sulfur melting route or in situ synthesis approach described earlier.

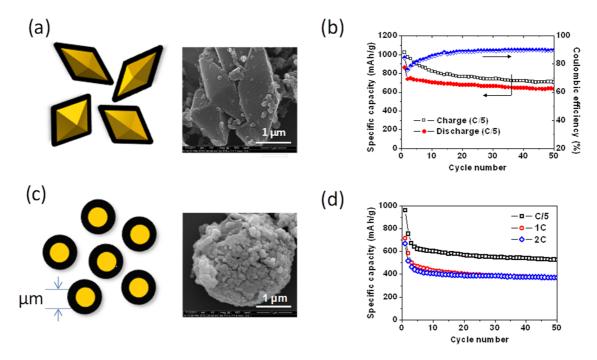


FIGURE 4. (a) Schematic illustration and scanning electron microscopy (SEM) image of bipyramidal sulfur particles coated with a layer of conductive polymer nanospheres. Reprinted from ref 56. Copyright 2012 American Chemical Society. (b) Cyclability of the bipyramidal composite at *C*/5 rate. Reprinted from ref 56. Copyright 2012 American Chemical Society. (c) Schematic illustration and SEM image of spherical sulfur particles coated with a layer of conductive polymer. Reprinted from ref 57. Copyright 2012 Royal Society of Chemistry. (d) Cyclability of the spherical composite at various rates. Reprinted from ref 57. Copyright 2012 Royal Society of Chemistry.

3. Sulfur—Polymer Hybrid Cathode Materials

Polymers can play a significant role in rechargeable Li-S batteries. For example, the attachment of poly(ethylene glycol) onto sulfur-carbon composites, which provides a chemical gradient that retards diffusion of polysulfides out of the electrodes, has been noticed to improve the electrochemical performance. 13,47 Yang et al. 48 have reported that coating of mesoporous carbon/sulfur composites with a conducting polymer poly(3,4-ethylenedioxythiophene) poly(styrene sulfonate) (PEDOT:PSS) improves the electrochemical performance. Although conductive polymers are a good choice because they provide a matrix for electron transport, they have much lower electronic conductivity than carbon materials, which prevents them being a sole electron conductor within the sulfur electrode. The diverse synthesis of conductive polymers provides opportunities for morphology control and materials design. We survey below a few strategies exploring conductive polymers as a conductive matrix or barrier for blocking polysulfides that can improve the electrochemical performance of Li—S batteries.

A few sulfur—conductive polymer composites with various structures and morphologies have been developed. For example, granular and tubular polypyrrole have been used as a conductive matrix and sulfur has been introduced by

melting and forming a thin layer (nanoscale) of sulfur on the surface of polypyrrole. 49,50 This outer coating strategy can improve sulfur utilization due to its nanosized coating, facilitating ion and charge transport. This approach could not reduce the dissolution of polysulfides in liquid electrolyte, but polypyrrole could act as a conductive matrix for sulfur, reducing agglomeration of large solid particles within the composite electrodes. An improved strategy is coating the sulfur particles with a conductive polymer (e.g., polythiophene or polypyrrole), forming a core—shell structure (Figure 4a,c). 55-57 Sulfur and discharged polysulfides are expected to be confined within the polymer shell, leading to good cyclability, efficiency, and rate capability (Figure 4b,d). This approach utilizes commercial sulfur or sulfur formed in aqueous solution by the thiosulfate reduction reaction with acid and postpolymerization involving self-assembly of these polymers onto the surface of sulfur particles.^{56,57} However, the large (micrometer-sized) sulfur particles lead to low utilization of sulfur. New synthesis methods are needed to form nanosized sulfur particles within the conductive polymer shells that could improve the utilization of active material.

Additional functionalities could be introduced into the conductive polymers to improve the performance further.

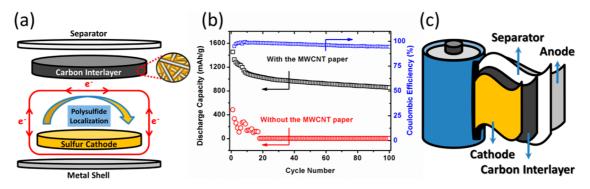


FIGURE 5. (a) Prototype Li—S coin cell configuration with a carbon interlayer in the cathode region. (b) Electrochemical performance of the Li—S cell with a MWCNT interlayer at 0.5 C rate. Reprinted from ref 9. Copyright 2012 Royal Society of Chemistry. (c) New Li—S cell configuration with a carbon interlayer.

For example, a mixed ionic—electronic conductor (MIEC) consisting of electronic and ionic conducting components^{58,59} could improve electron and ion transport, retard polysulfide dissolution, and maintain a robust but porous electrode structure improving the electrochemical performance. Future efforts could focus on multifunctional polymers or inorganic—organic hybrid composites with both ion and electron transport components as additives in the sulfur electrodes.

4. Novel Li-S Cell Configurations

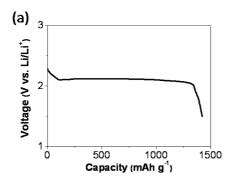
4.1. Conductive and Porous Carbon Interlayers. Recently, a novel Li—S cell configuration has been designed utilizing carbon interlayers between the separator and the regular sulfur electrodes for improving the cyclability of Li—S batteries (Figure 5a). Preliminary data show that a Li—S cell with an inserted conductive MWCNT paper exhibits outstanding cyclability with a capacity of >800 mA h g⁻¹ over 100 cycles (Figure 5b). The structure of the interlayer has to be porous or should possess many interspaces to accommodate the migrating active material during cycling. The dissolved polysulfides form a stable composite with the interlayer after charging/discharging, leading to excellent cyclability, long cycle life, and high Coulombic efficiency.

Current research spotlights on Li—S batteries are mostly on modifications "inside" sulfur cathodes. Although those approaches aiming at synthesizing composite materials or coating sulfur with protective layers improve the electrochemical performance, cost issues and challenges with large-scale manufacturing still remain as impediments for the commercialization of Li—S batteries. To overcome these problems, innovations "outside" sulfur electrodes could be a promising option, like the carbon interlayer configuration. Thus, conventional sulfur cathodes can be applied with a modification of Li—S cell configuration by inserting a conductive and porous carbon interlayer between the cathode

and separator (Figure 5c). This interlayer works as not only an "upper current collector" to improve the active material utilization but also a "polysulfide stockroom" to maintain the cycle stability and good system efficiency.

4.2. Li/Dissolved Polysulfide Cells. To enhance reaction kinetics in the electrodes, dissolved active material as catholytes are preferred like the electrolyte in conventional redox flow batteries and newly developed cathode-flow Liion batteries. 60,61 Lithium polysulfides are highly soluble in some organic solvents like tetrahydrofuran (THF). Up to 10 M polysulfide in THF can be prepared and a rechargeable lithium/dissolved sulfur battery has been demonstrated in 1979 by Rauh et al. 62 A capacity of 1,532 mA h g $^{-1}$ corresponding to 1.83 electrons per sulfur has been achieved in the first discharge, but limited cyclability was achieved with depths of discharge (DoD) as low as 0.5 electron per sulfur corresponding to a capacity of 418 mA h g^{-1} . Such cells show a single voltage plateau at \sim 2.1 V (Figure 6a). Recently, Zhang et al. 11 have modified the dissolved polysulfide electrolyte with LiNO₃ to promote Li anode passivation and improve Coulombic efficiency. The Li/dissolved polysulfide cells show superior performance to conventional Li-S cells in specific capacity and capacity retention. The predominant issue with these cells is the poor cycle life, 11,62 which is due to the passivation of lithium metal anode and cathode. Li metal anode needs to be improved by reducing lithium roughness, 63 and novel carbon cathode materials with engineered porosity and structure that can provide consistent electron transport while suppressing the passivation are needed to advance this approach to a practical stage.

The Li/dissolved polysulfide cell configuration could be further explored in a half-flow-mode Li—S battery with catholyte circulating at the cathode side like redox flow batteries (Figure 6b). A large amount of polysulfide dissolved in an external tank to the carbon current collector can result



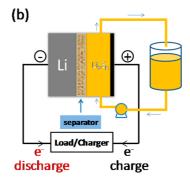


FIGURE 6. (a) First discharge voltage vs specific capacity profile of a Li/dissolved polysulfide cell at 2.8–1.5 V and C/10 rate and (b) a schematic of half-flow mode Li/dissolved polysulfide system.

in clogging the pores in the carbon electrodes due to the precipitation of high order polysulfides or sulfides upon cycling. This can be overcome by controlling the charge/discharge cutoff voltages to allow only soluble lithium polysulfides to be present in the catholyte, which would reduce the energy density of the overall systems but could lead to long cycle life. The half-flow mode Li—S batteries could be a promising candidate for large-scale storage for renewable energies.

5. Selection of Electrolyte Materials

Liquid electrolytes containing lithium salts (e.g., LiClO₄, LiCF₃SO₃, or Li(CF₃SO₂)₂N) and a mixture of cyclic or linear ethers (e.g., tetrahydrofuran, 1,3-dioxolane, dimethoxyethane, and tetra(ethylene glycol)—dimethyl ether) are commonly used in Li–S batteries. $^{63-65}$ Additives can have significant effect on Li–S performance by reducing the shuttle mechanism arising from the dissolution of polysulfides in the liquid electrolyte. Mikhaylik 66 discovered that additives with a N–O bond (e.g., LiNO₃) enhance passivation of lithium metal, forming a nonreactive surface exposed to dissolved polysulfides and thereby preventing the shuttle mechanism. This additive has been successfully applied in Li–S cells with almost 100% Coulombic efficiency with various cathode materials and cell configurations. $^{9-11,20,36,67,68}$

It is favorable to use lithium metal as anode in Li–S batteries since sulfur cathode is lithium-free and lithium has the highest capacity (ca. 3861 mA h g $^{-1}$). However, there are several problems with using lithium metal such as loss of lithium on each cycle, requiring excess Li that reduces the overall energy density, and reaction with polysulfides, resulting in insoluble Li₂S₂/Li₂S at the surface of lithium metal. In addition, the dendrite formation on the surface of lithium metal is a significant safety hazard. Use of a solid-state electrolyte that is impermeable to polysulfides can effectively suppress the formation of dendrites. For example,

Hassoun and Scrosati^{12,69} have developed a solid-state Li-S battery employing a poly(ethylene oxide)-type polymer electrolyte containing finely dispersed nanosized zirconia. High cell performance and Colombic efficiency were obtained at elevated temperatures. Hayashi et al. 70 have developed an all solid-state Li-S cell with Li₂S-P₂S₅ glassceramic electrolytes, which retained a capacity of over 650 mA h g^{-1} for 20 cycles at room temperature. Although high Coulombic efficiency can be achieved and no dendrite with lithium metal is formed with the all solid-state cell configuration, the low ionic conductivity of solid-state electrolyte limits their cell performance. Alternative solidstate electrolytes that can show high ionic conductivity $(\ge 0.5 \times 10^{-3} \text{ S cm}^{-1})$ and be stable with lithium metal are needed to further improve all solid-state Li-S batteries. Promising solid-state electrolytes include silicon-containing polymer electrolytes, 71,72 garnet-type lithium-ion conducting oxides, 73-76 and LISICON-type conductors. 77,78

6. Conclusions and Outlook

In this Account, we have surveyed recent developments on novel sulfur composite materials, new cell configurations, and electrolytes for rechargeable Li-S batteries. New nanostructured composites, facilitating rapid ion/electron transport while trapping polysulfides, have been largely developed. Sulfur-carbon nanocomposites have shown promising performances in terms of capacity and rate capability. Sulfurpolymer composites are also promising, and multifunctional polymers like MIEC could further improve the cell performance. Li-S cells with novel configurations like a carbon interlayer show impressive performance improvement. Li/dissolved polysulfide cells, an "old" configuration, hold the promise to regain renaissance if good improved lithium metal anodes and carbon cathodes can be developed. Flow mode Li/dissolved polysulfide cells are promising for grid energy storage applications. Electrolyte is equally important for solving the problems with Li-S batteries. The use of LiNO $_3$ additive solves the low-Coulombic-efficiency problem. New solid electrolytes with high ionic conductivity and stability with lithium metal anode are needed to make further advances.

In summary, significant improvements have been realized in the past decade with Li-S batteries, but challenges still remain. Development of novel nanocomposite materials with smart engineering design combined with new cell configurations will have a significant impact in the field. In addition, development of advanced materials and electrode characterization tools and methodologies can help enhance the fundamental understanding of the chemical and physical processes occurring. 79,80 The key issue with Li-S batteries to become a commercially viable technology is the poor reversibility of lithium anode in presence of the polysulfides dissolved in the electrolyte; this could become the limiting factor for Li–S cells over prolonged cycles, ca. 1000 cycles. Development of a selective coating layer on the lithium metal anode or a solid-state electrolyte, which allow only lithium-ion transport, could be an effective strategy to solve this problem. Li-S and Li-O₂ batteries are both appealing in terms of energy density, but the Li-O₂ system faces many more challenges. The poor cyclability and efficiency of Li-O2 cells along with the engineering challenges to keep the electrolyte and lithium metal free from contamination by ambient air make them far behind the Li-S batteries to become a practical technology. Rechargeable Li-S batteries are the most promising high energy density (>600 W h kg⁻¹) batteries for transportation and large-scale grid energy storage applications in the near future.

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FOOTNOTES

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REFERENCES

- 1 Manthiram, A.; Murugan, A. V.; Sarkar, A.; Muraliganth, T. Nanostructured Materials for Electrochemical Energy Storage and Conversion. *Energy Environ. Sci.* 2008, 1, 621–638.
- 2 Manthiram, A. Materials Challenges and Opportunities of Lithium Ion Batteries. J. Phys. Chem. Lett. 2011, 2, 176–184.
- 3 Bruce, P. G.; Scrosati, B.; Tarascon, J.-M. Nanomaterials for Rechargeable Lithium Batteries. Angew. Chem., Int. Ed. 2008, 47, 2930–2946.
- 4 Scrosati, B.; Garche, J. Lithium Batteries: Status, Prospects and Future. J. Power Sources 2010, 195, 2419–2430.
- 5 Goodenough, J. B.; Kim, Y. Challenges for Rechargeable Li Batteries. Chem. Mater. 2010, 22. 587–603.
- 6 Bruce, P. G.; Freunberger, S. A.; Hardwick, L. J.; Tarascon, J.-M. Li-O₂ and Li-S Batteries with High Energy Storage. *Nat. Mater.* 2012, *11*, 19–29.
- 7 Ji, X.; Nazar, L. F. Advances in Li-S Batteries. J. Mater. Chem. 2011, 20, 9821–9826.
- Mikhaylik, Y. V.; Akridge, J. R. Polysulfide Shuttle Study in the Li/S Battery System. J. Electrochem. Soc. 2004, 151, A1969

 –A1976.
- 9 Su, Y.-S.; Manthiram, A. A New Approach to Improve Cycle Performance of Rechargeable Lithium-Sulfur Batteries by Inserting a Free-Standing MWCNT Interlayer. *Chem. Commun.* 2012, 48, 8817–8819.
- 10 Su, Y.-S.; Manthiram, A. Lithium—Sulphur Batteries with a Microporous Carbon Paper as a Bifunctional Interlayer. Nat. Commun. 2012, in press.
- 11 Zhang, S. S.; Read, J. A. A New Direction for the Performance Improvement of Rechargeable Lithium/Sulfur Batteries. J. Power Sources 2012, 200, 77–82.
- 12 Hassoun, J.; Scrosati, B. Moving to a Solid-State Configuration: A Valid Approach to Making Lithium-Sulfur Batteries Viable for Practical Applications. Adv. Mater. 2010, 22, 5198— 5201.
- 13 Ji, X.; Lee, K. T.; Nazar, L. F. A Highly Ordered Nanostructured Carbon-Sulfur Cathode for Lithium-Sulfur Batteries. Nat. Mater. 2009, 8, 500–506.
- 14 Zhang, B.; Qin, X.; Li, G. R.; Gao, X. P. Enhancement of Long Stability of Sulfur Cathode by Encapsulating Sulfur into Micropores of Carbon Spheres. *Energy Environ. Sci.* 2010, 3, 1531–1537.
- 15 Schuster, J.; He, G.; Mandlmeier, B.; Yim, T.; Lee, K. T.; Bein, T.; Nazar, L. F. Spherical Ordered Mesoporous Carbon Nanoparticles with High Porosity for Lithium-Sulfur Batteries. *Angew. Chem., Int. Ed.* 2012, *51*, 3591–3595.
- 16 Jayaprakash, N.; Shen, J.; Moganty, S. S.; Corona, A.; Archer, L. A. Porous Hollow Carbon@Sulfur Composites for High-Power Lithium-Sulfur Batteries. *Angew. Chem., Int. Ed.* 2011, *50*, 5904–5908.
- 17 Ji, L.; Rao, M.; Zheng, H.; Zhang, L.; Li, Y.; Duan, W.; Guo, J.; Cairns, E. J.; Zhang, Y. Graphene Oxide as a Sulfur Immobilizer in High Performance Lithium/Sulfur Cells. *J. Am. Chem. Soc.* **2011**, *133*, 18522–18525.
- 18 Li, N.; Zhang, M.; Lu, H.; Hu, Z.; Shen, C.; Chang, X.; Ji, G.; Cao, J.; Shi, Y. High-Rate Lithium-Sulfur Batteries Promoted by Reduced Graphene Oxide Coating. *Chem. Commun.* 2012, 48, 4106–4108.
- 19 Ji, L. W.; Rao, M. M.; Aloni, S.; Wang, L.; Cairns, E. J.; Zhang, Y. G. Porous Carbon Nanofiber-Sulfur Composite Electrodes for Lithium/Sulfur Cells. *Energy Environ. Sci.* 2011, 4, 5053–5059.
- 20 Zheng, G.; Yang, Y.; Cha, J. J.; Hong, S. S.; Cui, Y. Hollow Carbon Nanofiber-Encapsulated Sulfur Cathodes for High Specific Capacity Rechargeable Lithium Batteries. *Nano Lett.* 2011, 11, 4462–4467.

- 21 Guo, J.; Xu, Y.; Wang, C. Sulfur-Impregnated Disordered Carbon Nanotubes Cathode for Lithium—Sulfur Batteries. Nano Lett. 2011, 11, 4288–4294.
- 22 Jeon, B. H.; Yeon, J. H.; Kim, K. M.; Chung, I. J. Preparation and Electrochemical Properties of Lithium-Sulfur Polymer Batteries. J. Power Sources 2002, 109, 89–97.
- 23 Cheon, S.-E.; Ko, K.-S.; Cho, J.-H.; Kim, S.-W.; Chin, E.-Y.; Kim, H.-T. Rechargeable Lithium Sulfur Battery II. Rate Capability and Cycle Characteristics. *J. Electrochem. Soc.* 2003, 150, A800–A805.
- 24 Han, S.-C.; Song, M.-S.; Lee, H.; Kim, H.-S.; Ahn, H.-J.; Lee, J.-Y. Effect of Multiwalled Carbon Nanotubes on Electrochemical Properties of Lithium/Sulfur Rechargeable Batteries. *J. Electrochem. Soc.* 2003, *150*, A889–A893.
- 25 Wang, J. L.; Yang, J.; Xie, J. Y.; Xu, N. X.; Li, Y. Sulfur-Carbon Nano-composite as Cathode for Rechargeable Lithium Battery Based on Gel Electrolyte. *Electrochem. Commun.* 2002, 4, 499–502.
- 26 Zhang, B.; Lai, C.; Zhou, Z.; Gao, X. P. Preparation and Electrochemical Properties of Sulfur-Acetylene Black Composites As Cathode Materials. *Electrochim. Acta* 2009, *54*, 3708–3713
- 27 He, G.; Ji, X.; Nazar, L. F. High "C" rate Li-S Cathodes: Sulfur Imbibed Bimodal Porous Carbons. Energy Environ. Sci. 2011, 4, 2878–2883.
- 28 Evers, S.; Nazar, L. F. Graphene-Enveloped Sulfur in a One Pot Reaction: A Cathode with Good Coulombic Efficiency and High Practical Sulfur Content. Chem. Commun. 2012, 48, 1233—1235.
- 29 Wang, C.; Chen, J.; Shi, Y.; Zheng, M.-S.; Dong, Q.-F. Preparation and Performance of a Core-Shell Carbon/Sulfur Material for Lithium/Sulfur Battery. *Electrochim. Acta* 2010, 55, 7010–7015.
- 30 Su, Y.-S.; Manthiram, A. A Facile In Situ Sulfur Deposition Route to Obtain Carbon-Wrapped Sulfur Composite Cathodes for Lithium-Sulfur Batteries. *Electrochim. Acta* 2012, 77, 272– 278.
- 31 Lee, K. P.; Chromey, N. C.; Culik, R.; Barnes, J. R.; Schneider, P. W. Toxicity of N-Methyl-2-Pyrrolidone (NMP) - Teratogenic, Subchronic, and 2-Year Inhalation Studies. *Fundam. Appl. Toxicol.* 1987, *9*, 222–235.
- 32 Elazari, R.; Salitra, G.; Garsuch, A.; Panchenko, A.; Aurbach, D. Sulfur-Impregnated Activated Carbon Fiber Cloth as a Binder-Free Cathode for Rechargeable Li-S Batteries. Adv. Mater. 2011, 23, 5641–5644.
- 33 Dorfler, S.; Hagen, M.; Althues, H.; Tubke, J.; Kaskel, S.; Hoffmann, M. J. High Capacity Vertical Aligned Carbon Nanotube/Sulfur Composite Cathodes for Lithium-Sulfur Batteries. Chem. Commun. 2012, 48, 4097–4099.
- 34 Wu, Z. C.; Chen, Z. H.; Du, X.; Logan, J. M.; Sippel, J.; Nikolou, M.; Kamaras, K.; Reynolds, J. R.; Tanner, D. B.; Hebard, A. F.; Rinzler, A. G. Transparent, Conductive Carbon Nanotube Films. *Science* **2004**, *305*, 1273–1276.
- 35 Ng, S. H.; Wang, J.; Guo, Z. P.; Wang, G. X.; Liu, H. K. Single Wall Carbon Nanotube Paper As Anode for Lithium-Ion Battery. Electrochim. Acta 2005, 51, 23–28.
- 36 Su, Y.-S.; Fu, Y.-Z.; Manthiram, A. Self-Weaving Sulfur-Carbon Composite Cathodes for High Rate Lithium-Sulfur Batteries. *Phys. Chem. Chem. Phys.* 2012, 14, 14495— 14499
- 37 Winn, D. A.; Shemilt, J. M.; Steele, B. C. H. Titanium Disulfide Solid-Solution Electrode for Sodium and Lithium. *Mater. Res. Bull.* **1976**, *11*, 559–566.
- 38 Whittingham, M. S. Chemistry of Intercalation Compounds Metal Guests in Chalcogenide Hosts. *Prog. Solid State Chem.* **1978**, *12*, 41–99.
- 39 Che, G.; Jirage, K. B.; Fisher, E. R.; Martin, C. R.; Yoneyama, H. Chemical-Vapor Deposition-Based Template Synthesis of Microtubular TiS₂ Battery Electrodes. *J. Electro-chem. Soc.* **1997**, *144*, 4296–4302.
- 40 Tsang, C.; Manthiram, A. Synthesis of Nanocrystalline VO₂ and Its Electrode Behavior in Lithium Batteries. J. Electrochem. Soc. 1997, 144, 520–524.
- 41 Lee, E. S.; Manthiram, A. High Capacity Li[Li_{0.2}Mn_{0.54}Ni_{0.13}Co_{0.13}]O₂—VO₂(B) Composite Cathodes with Controlled Irreversible Capacity Loss for Lithium-Ion Batteries. *J. Electro-chem. Soc.* **2011**, *158*, A47–A50.
- 42 Zaghib, K.; Simoneeau, M.; Armand, M.; Gauthier, M. Electrochemical Study of Li₄Ti₅O₁₂ as Negative Electrode for Li-lon Polymer Rechargeable Batteries. *J. Power Sources* **1999**, *81*, 300–305
- 43 Kavan, L.; Gratzel, M. Facile Synthesis of Nanocrystalline Li₄Ti₅O₁₂ (Spinel) Exhibiting Fast Li Insertion. Electrochem. Solid-State Lett. 2002, 5, A39–A42.
- 44 Huang, S. Y.; Kavan, L.; Exnar, I.; Gratzel, M. Rocking Chair Lithium Battery Based on Nanocrystalline TiO₂ (Anatase). *J. Electrochem. Soc.* **1995**, *142*, L142–L144.
- 45 Armstrong, A. R.; Armstrong, G.; Canales, J.; Garcia, R.; Bruce, P. G. Lithium-Ion Intercalation into TiO₂-B Nanowires. Adv. Mater. 2005, 17, 862–865.
- 46 Chen, H. Y.; Armand, M.; Courty, M.; Jiang, M.; Grey, C. P.; Dolhem, F.; Tarascon, J.-M.; Poizot, P. Lithium Salt of Tetrahydroxybenzoquinone: Toward the Development of a Sustainable Li-lon Battery. J. Am. Chem. Soc. 2009, 131, 8984–8988.
- 47 Wang, H.; Yang, Y.; Liang, Y.; Robinson, J. T.; Li, Y.; Jackson, A.; Cui, Y.; Dai, H. Graphene-Wrapped Sulfur Particles as a Rechargeable Lithium-Sulfur Battery Cathode Material with High Capacity and Cycling Stability. *Nano Lett.* 2011, 11, 2644–2647.

- 48 Yang, Y.; Yu, G.; Cha, J. J.; Wu, H.; Vosgueritchian, M.; Yao, Y.; Bao, Z.; Cui, Y. Improving the Performance of Lithium—Sulfur Batteries by Conductive Polymer Coating. ACS Nano 2011, 5, 9187–9193.
- 49 Liang, X.; Wen, Z.; Liu, Y.; Wang, X.; Zhang, H.; Wu, M.; Huang, L. Preparation and Characterization of Sulfur-Polypyrrole Composites with Controlled Morphology As High Capacity Cathode for Lithium Batteries. *Solid State Ionics* 2011, 192, 347–350.
- 50 Liang, X.; Liu, Y.; Wen, Z.; Huang, L.; Wang, X.; Zhang, H. A Nano-structured and Highly Ordered Polypyrrole-Sulfur Cathode for Lithium-Sulfur Batteries. *J. Power Sources* 2011, 196, 6951–6955.
- 51 Fu, Y.-Z.; Su, Y.-S.; Manthiram, A. Sulfur-Polypyrrole Composite Cathode Materials for Lithium-Sulfur Batteries. J. Electrochem. Soc. 2012, 159, A1420—A1424.
- 52 Zhang, S.-C.; Zhang, L.; Wang, W.-K.; Xue, W.-J. A Novel Cathode Material Based on Polyaniline Used for Lithium/Sulfur Secondary Battery. Synth. Met. 2010, 160, 2041– 2044.
- 53 Wang, J.; Yang, J.; Xie, J.; Xu, N. A Novel Conductive Polymer-Sulfur Composite Cathode Material for Rechargeable Lithium Batteries. Adv. Mater. 2002, 14, 963–965.
- 54 Sun, M.; Zhang, S.; Jiang, T.; Zhang, L.; Yu, J. Nano-wire Networks of Sulfur-Polypyrrole Composite Cathode Materials for Rechargeable Lithium Batteries. *Electrochem. Commun.* 2008, 10, 1819–1822.
- 55 Wu, F.; Chen, J.; Chen, R.; Wu, S.; Li, L.; Chen, S.; Zhao, T. Sulfur/Polythiophene with a Core/Shell Structure: Synthesis and Electrochemical Properties of the Cathode for Rechargeable Lithium Batteries. J. Phys. Chem. C 2011, 115, 6057–6063.
- 56 Fu, Y.-Z.; Manthiram, A. Orthorhombic Bipyramidal Sulfur Coated with Stacked Polypyrrole Nanospheres as Cathode Materials for Lithium-Sulfur Batteries. *J. Phys. Chem. C* 2012, 116. 8910–8915.
- 57 Fu, Y.-Z.; Manthiram, A. Core-Shell Structured Sulfur-Polypyrrole Composite Cathode Materials for Lithium-Sulfur Batteries. *RSC Adv.* **2012**, *2*, 5927–5929.
- 58 Murthy, A.; Manthiram, A. Highly Water-dispersible, Mixed Ionic-electronic Conducting, Polymer Acid-doped Polyanilines as Ionomers for Direct Methanol Fuel Cells. *Chem. Commun.* 2011, 47, 6882–6884.
- 59 Fu, Y.-Z.; Manthiram, A. Enhanced Cyclability of Lithium—Sulfur Batteries by a Polymer Acid-Doped Polypyrrole Mixed Ionic—Electronic Conductor. *Chem. Mater.* 2012, 24, 3081–3087.
- 60 Leon, C. P.; Frias-Ferrer, A.; Gonzalez-Garcia, J.; Szanto, D. A.; Walsh, F. C. Redox Flow Cells for Energy Conversion. J. Power Sources 2006, 160, 716–732.
- 61 Lu, Y.; Goodenough, J. B. Rechargeable Alkali-Ion Cathode-Flow Battery. *J. Mater. Chem.* **2011**, *21*, 10113–10117.
- 62 Rauh, R. D.; Abraham, K. M.; Pearson, G. F.; Surprenant, J. K.; Brummer, S. B. A Lithium/ Dissolved Sulfur Battery with an Organic Electrolyte. *J. Electrochem. Soc.* 1979, 126, 523–527
- 63 Mikhaylik, Y. V.; Kovalev, I.; Schock, R.; Kumaresan, K.; Xu, J.; Affinito, J. High Energy Rechargeable Li-S Cells for EV Application. Status, Remaining Problems and Solutions. ECS Trans. 2010, 25, 23–24.
- 64 Rauh, R. D.; Shuker, F. S.; Marston, J. M.; Brummer, S. B. Formation of Lithium Polysulfides in Aprotic Media. *J. Inorg. Nucl. Chem.* **1977**, *39*, 1761–1766.
- 65 Peled, E.; Stemberg, Y.; Gorenshtein, A.; Lavi, Y. Lithium-Sulfur Battery: Evaluation of Dioxolane-Based Electrolytes. J. Electrochem. Soc. 1989, 1621–1625.
- 66 Mikhaylik, Y. V. Electrolytes for Lithium Sulfur Cells. U.S. Patent 7,354,680, 2008.
- 67 Liang, X.; Wen, Z.; Liu, Y.; Wu, M.; Jin, J.; Zhang, H.; Wu, X. Improved Cycling Performances of Lithium Sulfur Batteries with LiNO₃-Modified Electrolyte. *J. Power Sources* 2011, 196, 9839–9843.
- 68 Aurbach, D.; Pollak, E.; Elazari, R.; Salitra, G.; Kelley, C. S.; Affinito, J. On the Surface Chemical Aspects of Very High Energy Density, Rechargeable Li-Sulfur Batteries. *J. Electrochem. Soc.* 2009, *156*, A694–A702.
- 69 Hassoun, J.; Scrosati, B. A High-Performance Polymer Tin Sulfur Lithium Ion Battery. Angew. Chem., Int. Ed. 2010, 49, 2371–2374.
- 70 Hayashi, A.; Ohtomo, T.; Mizuno, F.; Tadanaga, K.; Tatsumisago, M. All-Solid-State Li/S Batteries with Highly Conductive Glass-Ceramic Electrolytes. *Electrochem. Commun.* 2003, 5. 701–705.
- 71 Rossi, N. A.; West, R. Silicon-Containing Liquid Polymer Electrolytes for Application in Lithium Ion Batteries. *Polym. Int.* **2009**, *58*, 267–272.
- 72 Chinnam, P. R.; Wunder, S. L. Polyoctahedral Silsesquioxane—Nanoparticle Electrolytes for Lithium Batteries: POSS—Lithium Salts and POSS—PEGs. Chem. Mater. 2011, 23, 5111— 5121
- 73 Murugan, R.; Thangadurai, V.; Weppner, W. Fast Lithium Ion Conduction in Garnet-Type Li₇La₃Zr₂O₁₂. Angew. Chem., Int. Ed. 2007, 46, 7778–7781.
- 74 Ohta, S.; Kobayashi, T.; Asaoka, T. High Lithium Ionic Conductivity in the Gamet-Type Oxide $\text{Li}_{7-x}\text{La}_3(\text{Zr}_{2x}, \text{Nb}_x)\text{O}_{12}$ (x = 0-2). *J. Power Sources* **2011**, *196*, 3342-3345
- 75 Li, Y.; Wang, C.-A.; Xie, H.; Cheng, J.; Goodenough, J. B. High Lithium Ion Conduction in Garnet-Type Li₆La₃ZrTaO₁₂. *Electrochem. Commun.* **2011**, *13*, 1289–1292.

- 76 Xie, H.; Li, Y.; Goodenough, J. B. NASICON-type $\text{Li}_{1+2x}\text{Zr}_{2-x}\text{Ca}_x(\text{PO}_4)_3$ with High Ionic Conductivity at Room Temperature. *RSC Adv.* **2011**, *1*, 1728–1731.
- 77 Knauth, P. Inorganic Solid Li Ion Conductors: An Overview. Solid State Ionics 2009, 180, 911–916.
- 78 Minami, K.; Hayashi, A.; Tatsumisago, M. Electrical and Electrochemical Properties of the 70Li₂S · (30-x)P₂S₅ · xP₂O₅ Glass-Ceramic Electrolytes. *Solid State Ionics* **2008**, *179*, 1282–1285.
- 79 Gao, J.; Lowe, M. A.; Kiya, Y.; Abruna, H. D. Effects of Liquid Electrolytes on the Charge—Discharge Performance of Rechargeable Lithium/Sulfur Batteries: Electrochemical and in-Situ X-ray Absorption Spectroscopic Studies. *J. Phys. Chem. C* **2011**, *115*, 25132—25137.
- 80 Nelson, J.; Misra, S.; Yang, Y.; Jackson, A.; Liu, Y.; Wang, H.; Dai, H.; Andrews, J. C.; Cui, Y.; Toney, M. F. In Operando X-ray Diffraction and Transmission X-ray Microscopy of Lithium Sulfur Batteries. *J. Am. Chem. Soc.* **2012**, *134*, 6337–6343.